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Characterization of Bi$_2$Te$_3$ Nanostructure by Using a Cost Effective Chemical Solution Route

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ABSTRACT: An efficient and cost effective approach in the synthesis process of the bismuth telluride (Bi$_2$Te$_3$) powders and pellets were developed based on a chemical solution route. The route consists of dissolving of both the bismuth (III) nitrate pentahydrate, Bi(NO$_3$)$_3$.5H$_2$O and tellurium dioxide, TeO$_2$ into the same inorganic nitric acid, HNO$_3$ with the two–step precipitation of sodium hydroxide, NaOH and sodium borohydride, NaBH$_4$. The different characterization parameters such as X–Ray Diffraction (XRD), Scanning Electron Microscopy (SEM), Energy Dispersive X–ray (EDX), Transmission Electron Microscopy (TEM), Atomic Force Microscopy (AFM), UltraViolet (UV) absorbance and Fourier Transform InfraRed (FT–IR) spectrometry were carried out. As a result of these, the developed powders possessed a rhombo-hedral crystal structure exhibiting a nanocrystalline form with crystalline size about 10 nm. The elemental of Bi and Te were developed with their stoichiometric atomic ratio of (30.15):(48.19). Furthermore, the TEM micrographs showed an aggregate phenomenon and the primary crystalline size being quite low. Additionally, the produced Bi$_2$Te$_3$ pellets indicated a smooth surface with an average roughness value of 58 nm according to the AFM image. Absorption has occurred at about a range within 1 (arbitrary unit). Ultimately, the FTIR demonstrated that the C–H, O–H, C–O and C–S bonds were similar to the Bi$_2$Te$_3$ nanostructure materials.

KEYWORDS: Bi$_2$Te$_3$; Chemical solution route; Nanostructure materials; Nanocrystalline form.

INTRODUCTION

Bismuth telluride (Bi$_2$Te$_3$) is a semiconductor type material that is a compound of bismuth (Bi) and tellurium (Te). When the post–transition metal element of Bi alloys with a metalloid non–metal element of Te, it behaves like an acceptable ThermoElectric (TE) semiconductor material [1]. Recently, it has taken a great deal of fascinating attention from energy harvesting to chip cooling and sensing [2]. TE power generation devices perform a crucial role for utilization in the geothermal and solar power areas.  

TE generators and their parameter measurements have been presented some studies [3–5]. According to these reports, ThermoElectric Generators (TEGs) have less than
According to the authors’ previous review reports, a route has been presented for producing the efficient Bi$_2$Te$_3$ nanostructure. By means of the study, their implementations [19, 20] have been executed. The effect of organic and alkali modifiers on the crystal structure of any materials have been shown.

The main difference between this work and literature reports is that only one material has used for preparing the solution. For this reason, it has lots of advantages such as a sensitive control of the crystalline dimension, structure distribution and crystallinity of the end product of precursors. Also, it offers a significant enhancement for the chemical activities of the reactant, the possibility to replace the solid state method, and materials. Moreover, products of an intermediate state, a metastable state and a specific phase could be easily fulfilled by means of the process. Therefore, the proposed process is a cost effective synthesis technique. In order to complete the characterization of the nanostructure of Bi$_2$Te$_3$ powders and pellets produced by a cost effective chemical solution route, some of these studies have been demonstrated in this paper. Further research will be recommended to study the TE properties of Bi$_2$Te$_3$ nanostructures.

**EXPERIMENTAL SECTION**

**Developing of the nanostructure Bi$_2$Te$_3$**

The reagents of Bi(NO$_3$)$_3$.5H$_2$O ($\geq$98%, Sigma–Aldrich) and TeO$_2$ (≥97%, Sigma–Aldrich) were purchased form Sigma–Aldrich and used as starting materials for the co-precipitation of a chemical solution route. The solvents of HNO$_3$ (65%, Sigma–Aldrich), NaOH (98–100%, Sigma–Aldrich), NaBH$_4$ (98–100%, mark) and Ethanol (Analytical grade, Mark) were also purchased and used without further purification. These chemicals were weighted according to their stoichiometry (Bi$_2$Te$_3$) and prepared separate metal ion solutions by using 4 mmol (1.94 gm) Bi(NO$_3$)$_3$.5H$_2$O and 6 mmol (0.9576 gm) TeO$_2$. These were dissolved in the concentrated (2 M=31.86 ml) HNO$_3$. The NaOH, a stock solution of (3 M=30 gm) was employed for regulating pH value. In authors’ previous report [7], the developed procedure of the Bi$_2$Te$_3$ nanostructure was utilized. According to the procedure, firstly, Bi(NO$_3$)$_3$.5H$_2$O + 2 M HNO$_3$ and TeO$_2$ + 2 M HNO$_3$ were mixed together with solution 1. Then, the white precipitates (Solution 3)
were prepared with Solution 1 and 3M NaOH (Solution 2). After that, the black precipitates were generated with Solution 3 and NaBH₄ (Solution 4). Finally, the nanostructure Bi₂Te₃ powders and pellets were successfully developed by using a chemical solution route.

**Experimental technique**

Firstly, the XRD patterns were recorded by using a X’Pert high score PANalytical diffractometer with Cu–Kα radiation operated at 45 kV and 40 mA with angular range 10° ≤ 2θ ≥ 80°. Next, the morphology and atomic compositions (elemental) of the sample were also consummated with a SEM and an EDX of LEO 1430 VP systems. Then, the TEM measurement was performed for understanding the morphology changes of the synthesized Bi₂Te₃ nanostructure with the field emission TEM using JEOL JEM 2100F HRTEM microscope operated at 200 kV. After that, the 3D topographic surfaces were recorded in 0.01 nm discrimination using 10 MP CCD cameras by an AFM. In addition, the UV absorption was measured by using a UV–1800 recording spectrophotometer in the photon wavelength range between 200 and 800 nm. Finally, a Bruker Tensor II spectrometry was used for FTIR measurements.

**RESULTS AND DISCUSSIONS**

Simply by changing the reaction conditions, resides that a remarkable aspect of a chemical solution route versatility such as metals, oxides, sulphides and carbides can be prepared in various nanostructure forms. By using the bottom up approaches, it is based on the handling of the molecules and atoms to construct the compound TE materials. The formation of the white precipitate occurs from a homogeneous liquid phase because of a physical conversion and a chemical solution. In almost every case, the construction of advanced solid phase in a liquid medium simultaneously outcomes by nucleation and agglomeration of the particles [21]. In homogeneous solutions, the growth of nanosize particle can be regulated by controlling of the cations and anions. In addition, the kinetics of the precipitation and careful handling of the results can be developed for the monodisperse nanosize particles. By adjusting the circumstances, the particles with nanostructured distributions can be formed to determine the precipitation preceding such as the pH of solution and concentration of the reactants [22].

For the Bi₂Te₃ synthesis process, the co–precipitation involves with the precipitation of hydroxides by the addition of a NaOH solution indicating the solution of raw material. It is noted that the reaction chemistry applicable to the co–precipitation of Bi and Te described herein leads to the hypothesis that the white colour precursor is most likely a hydrated, very confidential mixture of Bi and Te. Thus, it demonstrated the advantage of control allowing over the stoichiometry of materials by using the homogeneous sample production and easily composite materials preparation.

Additionally, the metal solution is carried out by direct reaction between Bi(NO₃)₃.5H₂O and TeO₂ in the inorganic solution of HNO₃. It is reasonably studied that TeO₂ and Bi(NO₃)₃.5H₂O can be converted into Te⁵⁺ and Bi³⁺ by HNO₃. The expected chemical reaction is Bi⁺⁺ ions to associate with the reduced Te⁵⁺ ions to combine to form the Bi₂Te₃ during this chemical reaction as below:

\[
\text{Bi(NO}_3\text{)}_3\cdot5\text{H}_2\text{O (heat } \sim 70^\circ \text{C up to } 80^\circ \text{C)} \rightarrow 3\text{H}_2\text{O} + 2\text{HNO}_3 + \text{Bi(NO}_3\text{)}_3(\text{OH})_2 \\
2\text{TeO}_2 + \text{HNO}_3 \rightarrow \text{Te}_2(\text{NO}_3\text{)}_3\text{O}(\text{OH}) \\
\text{NaOH + HNO}_3 \rightarrow \text{NaNO}_3 + \text{H}_2\text{O} \\
2\text{Bi}_2\text{O}_3 + 9\text{TeO}_2 + 6\text{NaBH}_4 \rightarrow 2\text{Bi}_2\text{Te}_3 + 3\text{NaTeO}_3 + 6\text{H}_2\text{BO}_3 + 6\text{H}_2\text{O}
\]

This solution route has many advantages such as simple solution, faster producing, low–cost and utilization of the non–toxic materials. Furthermore, it allows the product to be free from other anions because herein it use the inorganic solutions having a low relative permittivity. Moreover, the production of TE materials at the molecular level offers a number of advantages by using the chemical solution route that can develop better homogeneity for multiphase materials and a cost effective bulk quantity production. Additionally, it allows the control of crystalline size and growth agglomeration at a molecular level. For these reasons, it might be possible to minimise the cost and efficiency for the production of the ultra–fine device quality Bi₂Te₃ nanopowders. According to previous report [23], this route has been monitored the best performance for producing the nanostructure powders when compared with thin films and electrochemical synthesis. Ultimately, both low toxicity and a cost effective solution are used by performing this route.
The advancement of these TE materials into next generation appliances crucially depends on understanding of the relationship between structure and characteristics. In this context, the XRD provides a quite well complimentary message about the nanostructure materials by showing the average coherence length as a function of direction. Its results give an overall view of the nanostructure form that is average over a large volume. The length of structural coherence between one and several tens of nanometers shows the nanostructure materials. A nanostructure material acts as a grating to a certain expansion. It produces the XRD pattern which shows a diffuse component and peaks. However, these peaks are observed in the XRD patterns of regular crystals. In addition, the diffuse components indicate a very strong with the non-crystalline structure. Fig. 1 illustrates the XRD spectra of the prepared Bi$_2$Te$_3$ powder materials.

The XRD spectra of the prepared sample demonstrated that the main reflections were quite well index to the reference code and matched to rhombohedra Bi$_2$Te$_3$ nanostructure (Reference Code 98–018–4631 which provides XRD machine for Bi$_2$Te$_3$ powder diffraction standards data same as JCPDS). There were not other crystalline impurities detected that indicating the phase purity of the Bi$_2$Te$_3$ powders. The spectrum showed that the structure indicated the nano crystalline with a (015) identified orientation (peak having sharp and higher intensity). When the spectrum with the Bi$_2$Te$_3$ nanostructure was compared the standard reference code, corresponding to (101), (015), (1010), (0111), (110) (205), (0210), (1115), (125), (2110) and (1211) planes, these were seen similar. These were in very good agreement with other researcher reports [24]. The broadening of the diffraction peaks demonstrated that the powders were a nanostructure form. The nanostructure crystallites were commonly observed as residing of the two structural distinct [25]. Especially, a remarkable fraction of atoms was characterized at the strained exterior for a single–crystal grain core and a surface shell layer. Moreover, the scattering centres produced some fringes that conclusively become "diffraction". At low interference, these fringes started off very broad range and at infinite interference it becomes indefinitely sharp. So, the determination of crystallite size was normally limited to the use of peak broadening. The strain was the other source of specimen broadening. If the crystallite is strained, then, the d spacing’s will be changed; a compressive stress would make the d spacing’s smaller, say reducing a given spacing d to d−δd.

The structural defects such as vacancies, interstitials, dislocations and layer faults induced inhomogeneous strain within crystalline materials. The degree of strain was obviously greater at distances closing to the actual defect. X–ray diffractogram provided a wealth of structural information about the nanomaterials. The micro and nanostructure materials were given by fundamental features such as the crystallite size, size distribution, the defect structure, texture, micro and macro strain, etc. The crystalline size and lattice train were estimated with well-known Scherrer equation [26] using the (015) reflection is ~9.25 nm and 0.0164, respectively, that confirmed reasonably well to the literature value [27].

The smallest crystallite material performed the lowest thermal conductivity and high resistivity. It had a crucial effect on the TE figure of merit, ZT. Recently, TE applications have attracted increasing interest due to its capability of converting

![XRD spectra of prepared Bi$_2$Te$_3$ powders.](image-url)
waste heat into electricity. In this overview, specifically the low dimensional materials are suitable for TE applications.

On the other hand, a SEM image was applied to determine the morphological and microstructural information about the prepared sample. Fig. 2 shows SEM images of the prepared Bi$_2$Te$_3$ nanostructure. Its homogeneity was arranged sequentially. This shows that the nanostructure is well agglomerated. It could be seen that the grains of Bi$_2$Te$_3$ exhibited initial powder morphologies, quasi spherical granule shapes in agglomerated clusters. The grain size were still in nanometer range according to the other researcher report [28].

Experiments suggested that the behaviour was related to grain boundaries. It indicated a stacking of nanosized particles and crystalline form with broad smooth surfaces to the crystallographic axis. Notably, an influence on the TE properties of this nanostructuration involved a bounding surface boundaries that were impeachable for broad phonon scattering circumstance. Also, the experimental results suggested that the microstructure of the scattering gave a critical consequence on the TE properties. Furthermore, the Bi$_2$Te$_3$ nanostructure would be able to achieve a high electrical conductivity and a power factor.

Fig. 2: SEM images and EDX spectrum of the prepared Bi$_2$Te$_3$ nanostructure.
In another measurement, an EDX was utilized to identify the atomic elemental composition of Bi₂Te₃ nanostructure. The EDX analysis was also crucial for improving accuracy of the quantitative compounds. Fig. 2b shows the EDX spectrum that materials of Bi and Te were arranged with their atomic stoichiometric ratio. The atomic composition of Bi and Te elements was approximately 2:3 (30.15):(48.19) within an instrumental accuracy. It was confirmed that the nanostructure was composed of only Bi and Te. The significant increase of an order of magnitude in figure of merit at low temperature was compared to the stoichiometric sample.

The Bi₂Te₃ is one of the promising candidates for TE applications due to its stoichiometric behaviour. However, its energy conversion performance is enhanced by its high thermal conductivity. The above result also shows the other peak which was carbon that indicated the small amount of contamination of the sample. For this reason, the H₂ gas was passed throughout the sample at ~250°C that the effects of post–depositional contamination. The oxygen containing functional groups were almost reduced in the procedure of reduction with NaBH₄. Thus, the metal ions were transformed into the Bi₂Te₃ nanostructure form. Eventually, the nanostructure material was enhanced for the TE power because of the quantum confinement. The size effect led to a carrier confinement. The selective scattering and the interface scattering was reduced. Additionally, the thermal conductivity was improved more than the electrical conductivity. Hence, the figure of merit can be increased in the Bi₂Te₃ nanostructure. On the other hand, contamination might be artificially produced, it occurs in the post–depositional environment. From these curves, it was also confirmed that the Bi₂Te₃ nanostructure was identified Bi and Te which was in good agreement with other researcher report [29]. A key feature of the microstructure was the homogeneous distribution of Bi the Te.

Fig. 3 shows the TEM micrographs of the Bi₂Te₃ nanostructure. The nanostructure exhibited an aggregate phenomenon, and the primary crystalline size was a low dimension. The mean crystalline size distribution was quite narrow. The crystals were connected via van-der-Waals forces, but not sintered together. Crystals in this dimension regime had about 50% of their atoms at the surface. From the XRD measurement, the investigation of the full width half maxima (FWHM) intensified in a crystallite size of ~9.25 nm. This was the best reconciliation with the crystalline size measured from the TEM image. In nanostructure materials, the effective scattering of acoustic phonons of the medium frequency with mean free path must be less than 100 nm. The contribution of phonons with mean free path between 5 and 100 nm has occurred about 55% of general scattering. Hence, the features could cause significant reduction in the thermal conductivity of TE behaviour.

About the crystallization, the roughness value of the manufactured pellets was observed by AFM studies that shown in Fig. 4. The AFM is developed to obtain a 3D image of a material surface on an ultramicroscopic scale. The AFM spectra showed that the sample had a nanostructure and uniform grains. There was not anomalous broad crystallization and the appearance of crystalline size was narrow. The surface texture was a crucial issue to interpret the nature of material surfaces. It was the random or repetitive deviation from the ordinary surface that forms the 3D topography of a surface. It plays a crucial role for the functional accomplishment of many scientific applications. It has a portrait that has shorter roughness properties are removed or filtered out and it also does not include any portrait transition due to transition in work piece geometry. So, it is significant to understand that it is always related to roughness when it comes to waviness. The waviness and average spacing between waviness peaks was determined from this profile. The waviness height which was the height from the top of the peak to the bottom of the trough was also defined from this profile. The average roughness height was usually at least three times. The surface roughness value is depends on the scale of measurement. Moreover, the concept of roughness value has a statistical implication that takes into the sample dimension and sampling interval.

Material with a narrow crystalline size was appreciated to maximize the potential for grain boundary. The nanostructure materials indicated a high strength and low ductility properties of the sample. It was recommended that a more superconductive exterior might help to advance the precursor decompositions. This facilitated the formation of exterior complexes that were well confirmed with other researcher report [30]. The average roughness value was about 58 nm which was
the mean value of the surface fluctuations relative to the centre plane of the sample. The low dimensional (~68 nm) surface roughness value was a strong repercussion for the TE behaviour of nanostructure materials.

The proper distribution of atom on the surface in nanostructure materials as well as their shape has a significant phenomenon on its characteristics. The results might sign that the grains were well arranged and therefore the prepared sample had the lowest scattering coefficient which leads to better conductivity. The boundary grains that could be clearly seen in the image and the results recommended that the electrical conductivity might be improved. The UltraViolet (UV) absorbance - visible absorption spectrum was recorded between 200 and 800 nm for the prepared sample. The absorbance spectrum is shown in Fig. 5 for the Bi\textsubscript{2}Te\textsubscript{3} nanostructure.

The exceptional absorption of the Bi\textsubscript{2}Te\textsubscript{3} nanostructure was found between 200 and 242 nm that might be due to the quantum incarceration consequence in nanostructure materials. The light moves to the blue end of the spectrum, as its wavelengths get shorter that represent the ‘blue shift’ formation. The blue shift might be caused by nano dimensional effect [31]. These values were calculated by using the simply Planck equation into the photon energy ranges between 5.1658 and 6.1990 eV, respectively. The shorter wavelengths of light were more energetic than longer wavelengths. The ultra violet
Fig. 4: AFM image of Bi$_2$Te$_3$ nanostructure pellet.

radiation had a low power of penetration. Especially in the range between 30 and 315 nm for nanostructure materials the direct and indirect energy gaps were observed. The similar absorption spectra of doped Bi$_2$Te$_3$ nanostructure were observed by the other researcher report [32]. The electron affinity activity was performing between 3.7 and 4.2 eV of the Bi$_2$Te$_3$ nanostructure.

The Fourier Transform InfraRed (FT-IR) spectrometry analysis process uses infra-red light to scan the test sample and investigate the chemical behaviour of the materials. Fig. 6 shows the FT-IR transmittance spectra of the Bi$_2$Te$_3$ nanostructure. The chemical bonds had characteristic frequencies at which they vibrate. They could be set on vibrations by illuminating the sample with infra-red light at the right frequency.

It can be seen from the Fig. 6 that corresponding to peaks at 2976.97 cm$^{-1}$, 2662.93 cm$^{-1}$, 1392.75 cm$^{-1}$ and 634.10 cm$^{-1}$ were assigned to C–H, O–H and C–S stretching, respectively on the prepared sample. The peaks at 2326.01 cm$^{-1}$, 2115.34 cm$^{-1}$ and 1992.54 cm$^{-1}$ were assigned to triple bond regime of C=C and C=N. Other identified major peaks being at 1265.85 cm$^{-1}$, 949.47 cm$^{-1}$ and 493.43 cm$^{-1}$ were assigned to the finger print. The FTIR spectrums comparing with the reference Bi$_2$Te$_3$ nanostructure showed that the C–H, O–H and C–S stretching were well matched [33]. The C–H stretching vibrations of the small close cage molecules occurred at a very high frequencies between 3160 and 3100 cm$^{-1}$, and their low intensities suggested that these bonds were almost homo polar. The C-H bond strengths were following the sequence of lower to higher order. The expected trend of larger vibration shows the correlation between the B-H stretching frequencies and the coupling constants. The most significant feature for the observed
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Fig. 5: UV spectrum of the prepared Bi$_2$Te$_3$ nanostructure.

(a) FT-IR spectrum

(b) Compare the result with a standard reference

Fig. 6: FT-IR spectra of the Bi$_2$Te$_3$ nanostructure.

CONCLUSIONS

In summary, a simple two-step co-precipitation chemical solution route was developed and appropriately established to synthesize the Bi$_2$Te$_3$. The precursor was reduced by NaBH$_4$ to produce a fine-particle. The experimental results revealed that the sample exhibited the nanostructure form of crystalline size of about 10 nm. This process was easy, adequate, less precarious and acceptable correlates to the other route. The characterizing aspects of the performed procedure were satisfactory for developing the Bi$_2$Te$_3$ nanostructure. Finally, many device quality Bi$_2$Te$_3$ nanostructure materials would be developed by using this procedure. It could be easily applicable on TE applications. This research work might open up a way for exploring high-performance of TE materials.

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